

Structural phase transitions in VSe₂: energetics, electronic structure and magnetism[†]

Georgy V. Pushkarev,^a Vladimir G. Mazurenko,^a Vladimir V. Mazurenko^a and Danil W. Boukhvalov^{*a,b}

First principles calculations of magnetic and electronic properties of VSe₂ describing the transition between two structural phases (H,T) were performed. Results of the calculations evidence rather low energy barrier (0.60 eV for monolayer) for transition between the phases. The energy required for the deviation of Se atom or whole layer of selenium atoms on a small angle up to 10° from initial positions is also rather low, 0.32 and 0.19 eV/Se, respectively. The changes in band structure of VSe₂ caused by these motions of Se atoms should be taken into account for analysis of the experimental data. Simulations of the strain effects suggest that the experimentally observed T phase of VSe₂ monolayer is the ground state due a substrate-induced strain. Calculations of the difference in total energies of ferromagnetic and antiferromagnetic configurations evidence that the ferromagnetic configuration is the ground state of the system for all stable and intermediate atomic structures. Calculated phonon dispersions suggest visible influence of magnetic configurations on vibrational properties.

1 INTRODUCTION

Monolayer VSe₂ is the one of the most intriguing members of the family of two-dimensional (2D) transition-metal dichalcogenides. This material attracts a special interest of the scientific community due to several recent discoveries, including in-plane piezoelectricity¹, a pseudogap with Fermi arc² at temperatures above the charge density wave transition (220 K for the monolayer³), and especially the existence of ferromagnetism in 2D system^{4–11}. Experimental results are rather contradictory. A strong room-temperature ferromagnetism with a huge magnetic moment per formula unit has been reported for monolayer VSe₂ epitaxially grown on graphite⁴. A local magnetic phase contrast has also been observed by magnetic force microscopy at the room temperature at the edges of VSe₂ flakes exfoliated from a three-dimensional crystal.¹² XMCD measurements evidence a spin-frustrated magnetic structure in VSe₂ on graphite.¹³ Paramagnetism of bulk VSe₂^{14,15} makes these observations more intriguing. Another situation was reported for the monolayers grown on bilayer graphene/silicon carbide substrate. In both works the absence of exchange splitting of the vanadium 3d bands observed in angle-resolved photoemission spectroscopy experiments was reported. This result contradicts to other studies that revealed a magnetization value not higher than 5 μ_B .^{16,17} Based on these results we can conclude that the influence of the substrate is im-

portant for description of the magnetic properties of these materials. Theoretical models have been developed to account for the above discrepant observations^{4,12,16,18}. These works mainly focused on the band structure and magnetic moments on vanadium sites. It has been proposed that the presence of charge density waves could cause the quenching of monolayer ferromagnetism due to the band gap opening induced by Peierls distortion¹⁹. Phonon spectra of several VSe₂ and similar systems were also considered theoretically^{20,21}. This modeling motivates us to study interplay between magnetism and structural phase transitions in VSe₂. Additionally, there is a plethora of works demonstrating a relationship between the symmetry, electronic structure and magnetic properties in transitional metal compounds^{22–25}.

The VSe₂ crystal is formed from separate layers along the c-axis direction. Two main phases for this material were predicted to be stable: the H phase characterized by Se stacked over each other and the T phase with one layers of Se rotated by 60° around axis normal to the plane of layer.¹⁸ Atomic structures of the VSe₂ monolayer in both H and T phases are shown in Fig.1. Surprisingly, the reported binding energies for different configurations are almost the same despite the colossal difference in magnetic properties and electronic structure (Fig.1).¹⁸ This finding additionally motivates us to examine various aspects of structural phase transitions in bulk, few-layer and monolayer of VSe₂.

^a Ural Federal University 620002, 19 Mira street, Ekaterinburg, Russia. Tel: +7 343 375 4444; E-mail: contact@urfu.ru

^b College of Science, Institute of Materials Physics and Chemistry, Nanjing Forestry University, Nanjing 210037, PR China

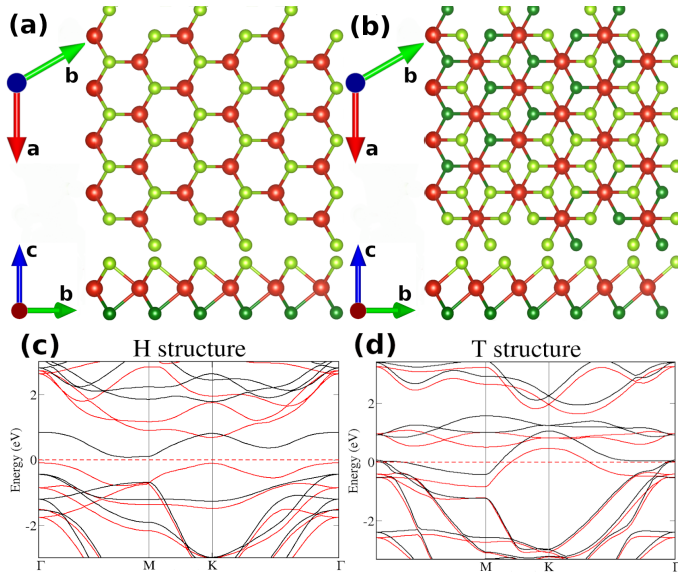


Fig. 1 Atomic structure of 2D VSe₂ monolayer (top and side view) in H phase (a) and in T phase (b). Vanadium atoms are denoted with red circles, upper and bottom selenium layers denoted with light green and dark green circles, respectively. (c) and (d) panels represent the corresponding spin-polarized band structures. Red lines correspond to spin up states and black ones to spin down, the Fermi level corresponds to 0 eV.

2 Computational method and model

Electronic properties of the VSe₂ system were simulated within Density Functional Theory (DFT) framework using the Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional²⁶ as implemented in the Vienna ab-initio simulation package (VASP)^{27,28} with a plane-wave basis set. This approach gave reliable results for other systems similar to VSe₂²⁹. Also we include van der Waals interaction using the method of Grimme (DFT (PBE)-D2)³⁰. Taking into account London dispersion forces is essential for few-layer VSe₂ (see Table 1 and discussion in section 3.5).

The calculation parameters were chosen as follows. The energy cutoff equals to 400 eV and the energy convergence criteria is 10^{-6} eV. For the Brillouin zone integration a $10 \times 10 \times 1$ gamma centered grid was used for layered structures and $8 \times 8 \times 8$ for bulk structures. A vacuum space more than 10 Å in the vertical z direction was introduced for layered structures. The technical parameters are similar to those used in the recent studies of phase stability in layered systems.^{31,32}

The optimized atomic positions for T-phase and lattice parameters $a = b = 3.31$ Å and $c = 6.20$ Å are in good agreement with experiment³³. In particular, the corresponding interlayer distance in bulk VSe₂ is 3.04 Å. The calculated band structures of VSe₂ monolayer in the T and H phases are in good agreement with previous works.¹⁸ The calculated magnetic moment of $0.68 \mu_B$ for initial configuration without rotation of the selenium atoms also agrees with results of the previous work³⁴.

To investigate the transition between H and T phases we performed self-consistent calculations of electronic structure and to-

tal energies in transitional points between these phases. For this purpose, we rotate either one Se atom or all selenium atoms belonging to the upper layer of VSe₂ in supercell as schematically shown in Fig.2. To trace the changes in electronic structure and magnetic properties the calculations for configurations with a 10° rotation step were performed. Generally, the rotation can be realized within two models. The first one is to move Se in plane from initial to final point (Fig.2 a and c). The second one is to fix the constant V-Se distance for all intermediate steps, which produces an elevation of selenium atoms above the plane at intermediate steps of the migration (Fig.2 b and d). We will refer these rotation models as in-plane and arc rotation schemes, respectively. All the calculations were performed for the ferromagnetic ordering of the spins of vanadium atoms.

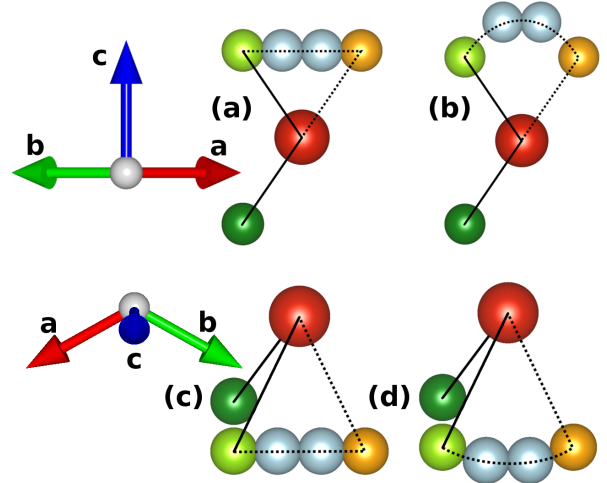


Fig. 2 Schematic visualization of the plane (a,c) and arc (b,d) types of the Se atoms rotation. (a,b) and (c,d) panels correspond to side and top views, respectively. Initial and final positions of Se are presented with orange and green circles, respectively. Intermediate configurations of selenium atoms obtained with the 20° step are denoted with light blue circles.

3 Results and discussion

3.1 Rotation of single Se atom

At the first step of our study we have simulated the motion of the single Se atom in the monolayer (see Fig. 3). For simplicity, we considered an in-plane migration of the atom. Results of the calculations (Fig. 3) evidence a gradual increasing of the total energy of the system during the all processes of the rotation with maximal value at final point. The cause of the large magnitude of the energies and instability of the final configuration is in decreasing of the distance between moved and rigid Se atoms to the value of 1.92 Å. Thus we can conclude that the model of the single Se atom rotation is unrealistic and transition between T and H phases may be realized only with distortion of the whole selenium layer. Further we will consider only this kind of the structural phase transitions. The values of the magnetic moments calculated for intermediate configurations (Fig. 3) support our initial guess that the structural transition between the phases affects magnetic properties of VSe₂. Note that a deviation of the

selenium atoms from equilibrium positions on small angles (less than 10°) requires much smaller energies of about 0.32 eV and, therefore, should be taken into account for a realistic description of the atomic structure of VSe_2 at the room temperature.

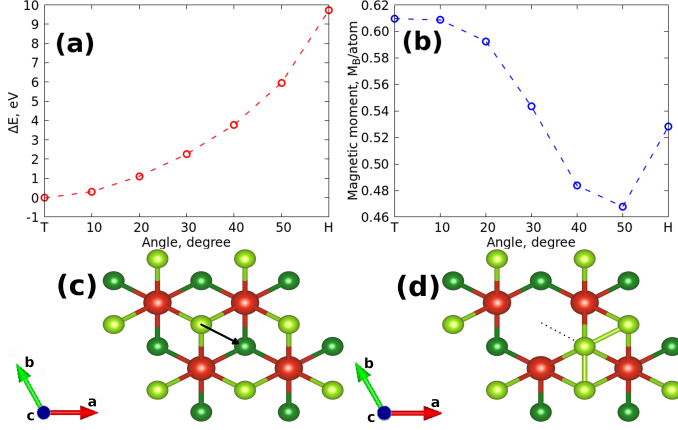


Fig. 3 Evolution of the total energy (a) and magnetic moment (b) during in-plane rotation of single Se atom. (c) and (d) panels visualize the initial and final atomic structures. Light and dark green circles denote upper and bottom selenium layers, respectively.

3.2 Rotation of the whole Se sheet in the VSe_2 monolayer

Having considered the results concerning the migration of the single Se atom we are in a position to analyze the case of whole upper Se-layer rotation, which will provide a better understanding of the transition between H and T phases.

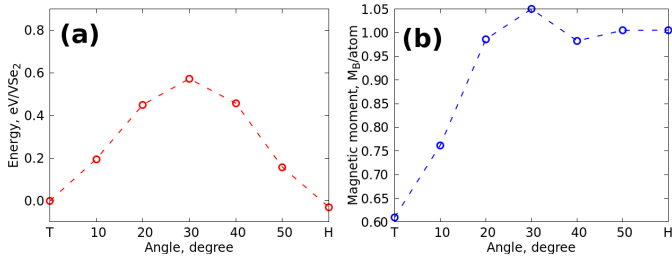


Fig. 4 Evolution of the total energy (a) and magnetic moment (b) during rotation of whole upper Se layer of VSe_2 monolayer within arc model.

The performed simulations for 3×3 supercell with constant V-Se distances when Se atoms elevate from initial and starting positions (Fig. 2) revealed the energy barrier of 0.60 eV that is smaller than that observed in the case of the in-plane rotation (Fig.S1a in SI). Thus further we will consider only this type of the Se atom migration. To evaluate the temperature required to overcome this barrier one should establish a relation between the calculated energies of the process and temperature of the reactions. We have addressed this question in our previous work³⁵ and found that the barrier values of about 0.50 eV and 1.20 eV correspond to the room temperature and 200 °C, respectively. Thus, the energy barrier of 0.60 eV can be overcome already at the temperatures about 40 °C.

Four conclusions could be drawn from these results. (i) There is a possibility of the structural phase transition in previously stud-

ied VSe_2 samples during measurements. (ii) For development of devices based on VSe_2 and similar monolayer systems one should take into account possibility of the structural phase transitions caused by the heating of the devices during work. Such a transition can significantly affect the work of the device due to difference in electronic structures of different phases (see Fig. 1, also changes in band structure Fig.S3 in SI). (iii) One can use VSe_2 and similar systems as temperature detectors. (iv) According to our results there is a low-energy cost to deviate the selenium atoms belonging to one layer on a small angle from the equilibrium positions. It means that one needs to account this for a realistic interpretation of the experimental data.

Moderate temperature of the transition between different structural phases requires an examination of the electronic structure and magnetic properties at intermediate steps of the structural phase transition. The obtained calculations results demonstrate that in the case of the ferromagnetic ground state the values of magnetic moments change gradually with small step of 10° of the rotation of Se layer. From Fig.4 one can see that at 30° the magnetic moment has the maximal value of $1.05\mu_B$, which is about two times larger than that in the initial configuration. According to the calculated occupation matrices such a magnetic moment change is mainly related to the contributions of xy and $x^2 - y^2$ orbitals of vanadium atoms (see Fig. 5). Since the total occupation (spin-up + spin-down) of the different orbitals remain almost the same, the orbital magnetic moment values change is fully connected with a redistribution of the electrons between different spin channels due to change of the hybridization between V and Se.

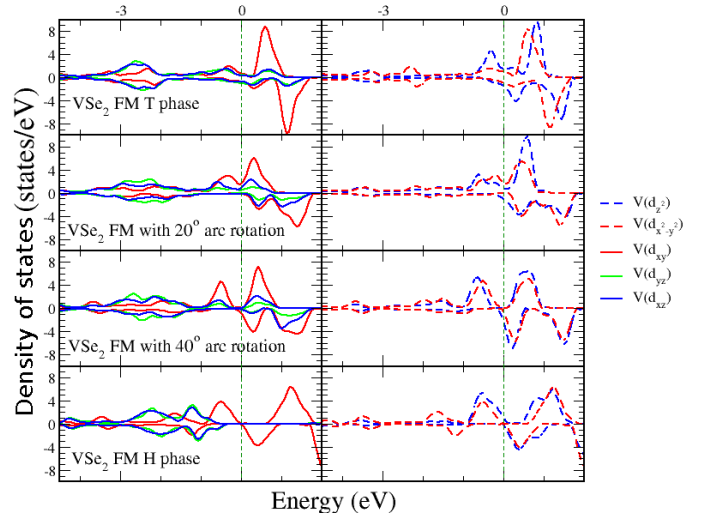


Fig. 5 Partial densities of states calculated for VSe_2 monolayer in the ferromagnetic configuration. The arc rotation scheme with the 20° step was used. Left and right panels correspond to (d_{xy}, d_{yz}, d_{xz}) and $(d_{x^2-y^2}, d_{x^2-y^2})$ sets of states, respectively.

In the case of an antiferromagnetic configuration the situation is more complicated. First of all, the magnetic lattice of VSe_2 is frustrated one, which is in agreement with experimental observations.¹³ This means that within a mean-field DFT approach we cannot define an antiferromagnetic collinear-type order cor-

responding to minimum of the magnetic interaction energy for all V-V bonds, simultaneously. The second complication follows from the fact that the system in question is metal. It means that the magnetization of individual vanadium atom can be very sensitive to the orientation of the neighbouring magnetic moments³⁶. Indeed, our DFT simulations of the VSe₂ supercell with antiferromagnetic ordering have revealed a strong suppression of the magnetic moment values of some vanadium atoms in the supercell. In addition, we observe that the details of the magnetic moments suppression strongly depend on the size of the supercell. In this complex situation some information on magnetic couplings in the VSe₂ system could be extracted by using the theory of infinitesimal spin rotations approximation^{36,37}. However, the magnetic couplings calculated in this way can be used for analysis only in the vicinity of the ferromagnetic configuration.

The values of the magnetic moments in AFM phase can be stabilized by inclusion of the on-site Coulomb interaction as can be done with DFT+*U* approach. However, the using of the DFT+*U* approach in the case of VSe₂ is questionable, since the experimental ARPES spectra are in good agreement with GGA band structure as it was shown in Refs.^{13,16,38}. At the same time the inclusion of the Hubbard *U* leads to considerable changes in the band structure.

Thus, the energy difference between AFM and FM solutions for VSe₂ simulated with GGA does not allow us to construct a comprehensive magnetic model and estimate the corresponding magnetic interactions between vanadium atoms. Nevertheless, the results of these calculations evidence that despite the changes of electronic structure at intermediate steps the ferromagnetic configuration remains significantly energetically favorable in all the cases (Fig.6). Thus the possible structural distortions in VSe₂ will not provide a suppression of ferromagnetism. Our calculations demonstrate that possible transition from experimentally observed T phase toward H phase should provide an enhancement of ferromagnetic interactions and increasing of magnetic moment. To simulate the experimentally observed paramagnetic state of bulk VSe₂^{14,15} one can use a dynamical mean-field theory.

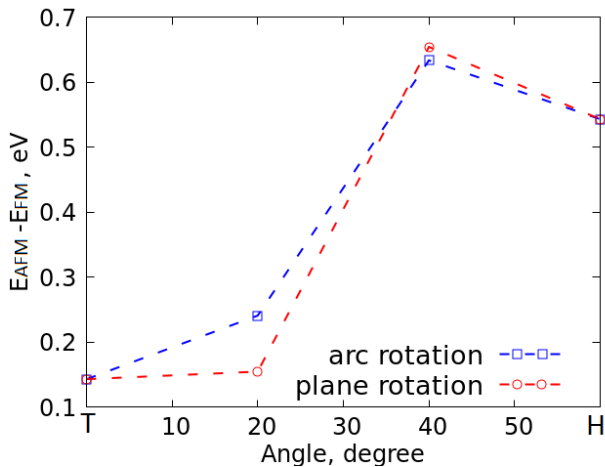


Fig. 6 Difference of AFM and FM state total energies calculated in arc and plane rotation schemes for 3x3 supercell of VSe₂ monolayer.

3.3 Structural phase transition in bulk VSe₂

There are two main differences in the energetics of the structural phases transitions in bulk and monolayer VSe₂. The first one is the almost the same value for the energies of the motion of Se layer within both rotation models (Fig.7 and Fig.S1c in SI). The second one is increasing of the migration barrier (see Fig. 7). Both are related to the van der Waals interactions between the layers in the bulk VSe₂. The analysis of the calculated partial density of states in this case leads to similar conclusions as above (see Fig.S2 in SI)

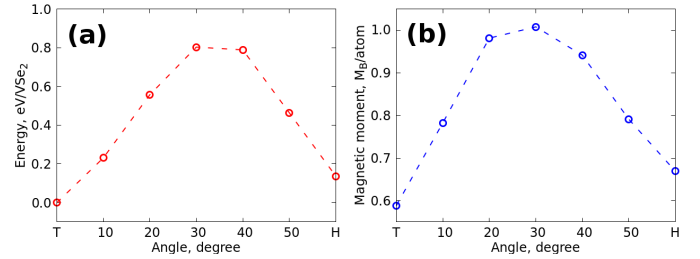


Fig. 7 Total energy (a) and magnetic moment (b) as functions of the rotation angles. The simulation results were obtained for bulk VSe₂ within the arc rotation model.

In the case of the rotation of the Se atoms belonging to one layer with constant V-Se distance at intermediate steps of migration, the initial distance of 3.63 Å between rotated and fixed selenium layers decreases by 0.54 Å. This deviation from the optimal interlayer distance provides an increasing of the energy barrier (see also changes in band structure Fig.S4 in SI). The value of the energy barrier is corresponding with stability of the structural ground state in bulk crystal up to the temperatures above 100°C. Note that in contrast to monolayer case the structural ground state of bulk VSe₂ is T configuration with ferromagnetic orientation of magnetic moments.

3.4 Structural phase transition in bi- and trilayers of VSe₂

Moreover, we examine the energetics of the structural phases transition in the top layer of bi- and trilayers VSe₂ with different stacking models (Fig. 8). The notation of the types of Bernal stacking is similar to graphite. These results also can be applied for VSe₂ non-covalently attached to substrates.

Results of the calculations (Fig. 9) evidence similarity the case of few-layer VSe₂ with monolayer. The configuration of the H type corresponds to the structural ground state for all types of the stacking in few-layer case. The energy required for the transition from T to H phase is about 0.60 eV for AA- and AB- stacking in bilayer. In trilayer the most energetically favorable stacking orders are AAA and ABC.

Thus, similarly to free standing VSe₂ monolayer, in top layer of VSe₂ there can be transition between two structural configurations at a moderate heating. The magnetic moments of vanadium atoms belonging to the upper layer of the few-layer structures change from 0.64 μ_B to 0.82 μ_B. Such a change is fully connected with a redistribution of the electrons between different spin channels, main contributions are from *xy* and *x*² - *y*² orbitals of vanadium atoms similar to monolayer case. Therefore, the presence

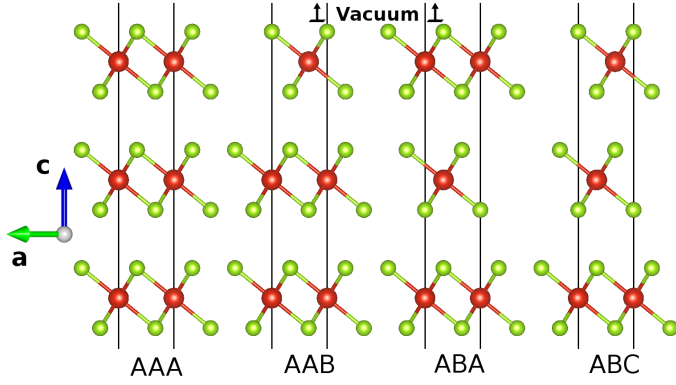


Fig. 8 Schematic representation of the unit cells used for simulating VSe₂ trilayers characterized by different stacking models.

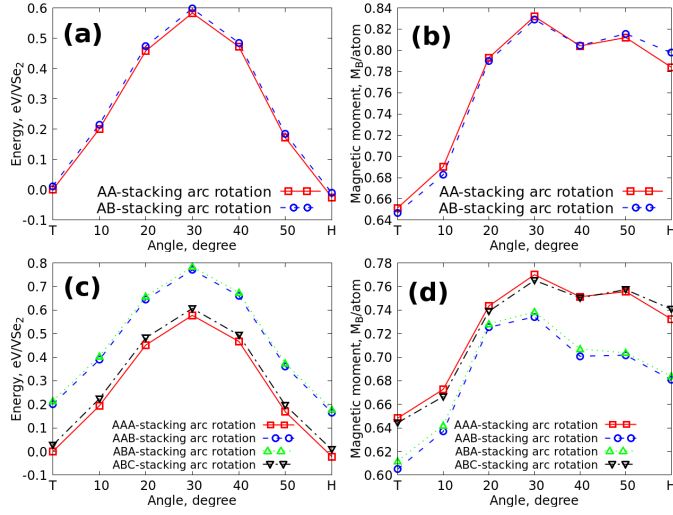


Fig. 9 Total energy (left panels) and magnetic moment (right panels) of two- (a,b) and three-layer (c,d) VSe₂ systems estimated for H, T and intermediate structures.

of the substrates does not influence significantly on sensitivity of bi- and trilayer VSe₂ systems to structural changes.

3.5 Interlayer binding energy

To understand the effect of interlayer interactions on structural properties we have checked interlayer distances and binding energies. The binding energies E_b for different VSe₂ structures were calculated by using the following expression $E_b = (E - n \cdot E_{\text{mono}}) / m$, where E is the total energy of considered system, E_{mono} - total energy of monolayer, n - number of layers in the considered system, m - average number of interlayer interactions ($m = 2, 3/2$ and 1 for bulk, 3 and 2-layers, respectively). Results of these calculations are presented in Table 1.

In the case when van der Waals interaction is neglected we obtain that the distance between V-V atoms belonging to the same layer is 3.33 Å and the Se-Se interlayer distance equals to 3.12 Å. When the van der Waals interaction is taken into account such distances equal to 3.31 Å and 3.04 Å, respectively. In 2- and 3-layer cases we considered the structures (Fig.8) with the lowest total energies. Calculated values of the binding energies evidence

that few-layer VSe₂ is pure van der Waals structure in contrast to bulk VSe₂ where London dispersion forces is a small addition to electrostatic interactions between V-cations and Se-anions from different layers. The changes of interlayer distances are proportional to contribution of the dispersion forces to the binding energies (about 0.1 Å in bulk and 0.4 - 0.6 Å in few-layer systems). Therefore, the energy difference in migration barriers in bulk and few-layer VSe₂ can be explained by contribution from electrostatic repulsion of anions from the layer above.

VSe ₂ structure	E_b with vdW, meV	E_b without vdW, meV	Interlayer distance with vdW (without vdW), Å
T-bulk	7.93	4.79	3.04(3.12)
H-bulk	99.67	95.78	3.22(3.32)
T-two	15.51	-9.36	3.11(3.51)
H-two	24.68	-29.86	3.69(4.27)
T-three	19.05	-90.86	3.08(3.57)
H-three	100.82	-51.77	3.66(4.14)

Table 1 Interlayer binding energies (meV/formula unit) and interlayer distances calculated for different VSe₂ structures with and without vdW interaction.

3.6 Phonon dispersion

To complete the picture of physical properties of VSe₂ monolayer we have performed calculations of phonon dispersions by using VASP and Phonopy packages³⁹. These combination of the packages is widely used for studying of vibrational properties in similar systems.³¹ For such calculations we used $3 \times 3 \times 1$ supercell to obtain sets of forces and mesh grids: $10 \times 10 \times 1$ for monolayer and $6 \times 6 \times 6$ for bulk. Both H and T phases in nonmagnetic and ferromagnetic configurations were considered.

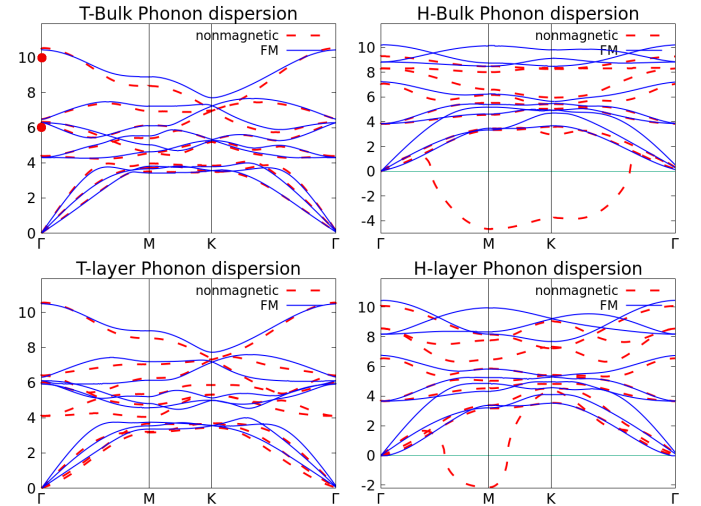


Fig. 10 Phonon dispersions calculated for nonmagnetic (red dashed line) and ferromagnetic state (blue solid line) of monolayer and bulk VSe₂. Both T and H phase structures are presented. Red dots denote experimental frequencies taken from Ref. 40.

The calculated phonon spectra are presented in Fig. 10. For the T phase systems (bulk and monolayer) the resulting dispersions

demonstrate a weak sensitivity to the magnetism. It is not the case for the H phase configurations. In the nonmagnetic state for H-monolayer and H-bulk we observe a soft phonon mode in the direction Γ –M–K for monolayer and in all symmetry directions for bulk. Existence of such a mode indicates structural instability. Importantly, in the ferromagnetic case the soft mode disappears, which means that the account of magnetism provides structural stability of H phase in both monolayer and bulk. The cause of this effect of magnetic configurations is the robustness of magnetic interactions (see Fig.6 and discussion above) which is the same order of magnitude as difference between structural phase. For H-bulk and T-monolayer ferromagnetic systems the calculations reveal the appearance of the indirect gap of 0.57 THz. Comparison of the calculated dispersion curves with available experimental data from Ref. ⁴⁰ obtained by a point-contact spectroscopy and Raman methods can be fulfilled only for the Γ point for which experimental oscillation frequencies are 6.04 (25 meV) and 9.67 (40 meV) THz. Our theoretical values of 6.28 and 10.42 THz are in good agreement with experimental data.

3.7 Structural phase transition by stretching

The last step of our survey is the modeling of stretch which can appear in the monolayer due to substrate influence. To simulate this effect, we increase a and b lattice vectors of our structure and then relax atomic positions to find a new ground state corresponding to new lattice parameters. Results of the calculation evidence that a stretching more than 3 percent leads to phase transition of the ground state configuration from H to T in monolayer and bilayer VSe₂ (Fig. 11a).

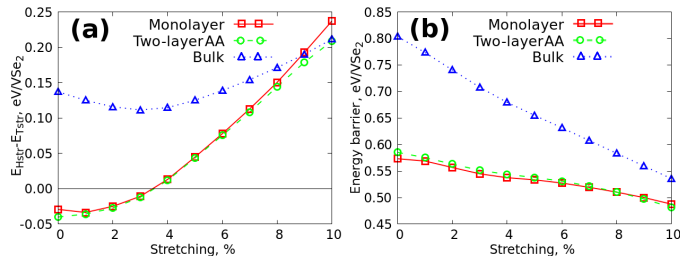


Fig. 11 Energy difference between H and T structures of VSe₂ (a) and energy barrier (b) as functions of stretching in a and b lattice directions. Lines of different colors correspond to systems with different numbers of layers.

Therefore, the experimentally observed structure¹³ of T type can result from a substrate-induced strain. Another effect of the stretching is a decreasing energy barrier for migration between different configurations (Fig. 11b). Here we define the energy barrier as energy difference between T structure and intermediate 30° structure

4 CONCLUSIONS

Results of first-principles calculations demonstrate that the energy barrier for the transition between two structural states of VSe₂ monolayer with a step-by-step rotation of the single Se atom is rather high. From the other hand the energy cost of the rotation of whole selenium layer is rather low (about 0.60 eV for

monolayer and 0.80 eV for bulk). In the case of the monolayer it could be realized with a heating of the samples. The excitation energies of the rotation of the selenium layer up to 10° are very low, therefore, the realistic theoretical description of VSe₂ (from monolayer to bulk) should take into account these small deviations from ideal crystal structure.

Our calculations demonstrate that the transition from the experimentally observed T configuration to the H configuration is accompanied by a considerable change in electronic structure which is a redistribution of 3d electrons of vanadium between orbitals. Such transitions significantly influence on transport and thermal properties of VSe₂. From the other hand, the values of magnetic moments and total energies of ferro- and antiferromagnetic configurations change gradually between two structural phases.

In all the considered cases (bulk, few-layer and monolayer) system demonstrate strong favorability of ferromagnetic structure. The analysis of the calculated phonon dispersions has demonstrated a principal role of the ferromagnetism in stabilization of the atomic structure of the VSe₂ monolayer in H phase and similar systems. On the basis of the obtained results we can conclude that the experimentally observed paramagnetism in bulk VSe₂ and contradictory results of magnetic measurements for monolayers on different substrates are not caused by structural changes.

The calculations for bi- and trilayers demonstrate that the energy barrier of transition is similar to monolayer. The strain, possibly induced by the substrate, provides the change of the most energetically favorable structure from H to T. Therefore, the experimental observation of T configuration can result from a VSe₂ structure stretching by more than 3 percent on substrates. Another effect of the stretching is a decrease of the energy barrier of transition between structural phases. Thus both strain and deviation from ideal structure should be taken into account for realistic description of VSe₂ monolayer on substrates.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the Russian Science Foundation, Grant No. 18-12-00185.

Notes and references

- 1 J. Yang, A. Wang, S. Zhang, J. Liu, Z. Zhong and L. Chen, *Phys. Chem. Chem. Phys.*, 2019, **21**, 132–136.
- 2 Y. Umemoto, K. Sugawara, Y. Nakata, T. Takahashi and T. Sato, *Nano Research*, 2019, **12**, 165–169.
- 3 P. Chen, W. Pai, Y. Chan, V. Madhavan, M. Chou, S. Mo, A. Fedorov and T. Chiang, *Physical Review Letters*, 2018, **121**, 196402.
- 4 M. Bonilla, S. Kolekar, Y. Ma, H. Coy Diaz, V. Kalappattil, R. Das, T. Eggers, H. Rodriguez Gutierrez, M.-H. Phan and M. Batzill, *Nature Nanotechnology*, 2018, **13**, 289–293.
- 5 Q. Wu, Y. Zhang, Q. Zhou, J. Wang and X. C. Zeng, *The Journal of Physical Chemistry Letters*, 2018, **9**, 4260–4266.

- 6 Z. Wang, T. Zhang, M. Ding, B. Dong, Y. Li, M.-L. Chen, X. Li, J.-Q. Huang, H. Wang, X. Zhao, Y. Li, D. Li, C. Jia, L. Sun, H. Guo, Y. Ye, D. Sun, Y. Chen, T. Yang and Z. Zhang, *Nature Nanotechnology*, 2018, **13**, pages554–559.
- 7 D. J. O'Hara, T. Zhu, A. H. Trout, A. S. Ahmed, Y. K. Luo, C. H. Lee, M. R. Brenner, S. Rajan, J. A. Gupta, D. W. McComb and R. K. Kawakami, *Nano Letters*, 2018, **18**, 3125–3131.
- 8 S. Jiang, L. Li, Z. Wang, K. F. Mak and J. J. Shan, *Nature nanotechnology*, 2018, **13**, 549–553.
- 9 N. C. Frey, H. Kumar, B. Anasori, Y. Gogotsi and V. B. Shenoy, *ACS Nano*, 2018, **12**, 6319–6325.
- 10 D. V. Averyanov, I. S. Sokolov, A. M. Tokmachev, O. E. Parfenov, I. A. Karateev, A. N. Taldenkov and V. G. Storchak, *ACS Applied Materials & Interfaces*, 2018, **10**, 20767–20774.
- 11 N. Samarth, *Nature*, 2017, **546**, 216–218.
- 12 S. Lee, J. Kim, Y. C. Park and S.-H. Chun, *Nanoscale*, 2019, **11**, 431–436.
- 13 P. K. J. Wong, W. Zhang, F. Bussolotti, X. Yin, T. S. Herng, L. Zhang, Y. L. Huang, G. Vinai, S. Krishnamurthi, D. W. Bukhvalov, Y. J. Zheng, R. Chua, A. T. N'Diaye, S. A. Morton, C.-Y. Yang, K.-H. Ou Yang, P. Torelli, W. Chen, K. E. J. Goh, J. Ding, M.-T. Lin, G. Brocks, M. P. de Jong, A. H. Castro Neto and A. T. S. Wee, *Advanced Materials*, 2019, **0**, 1901185.
- 14 C. van Bruggen and C. Haas, *Solid State Communications*, 1976, **20**, 251 – 254.
- 15 M. Bayard and M. Sienko, *Journal of Solid State Chemistry*, 1976, **19**, 325 – 329.
- 16 J. Feng, D. Biswas, A. Rajan, M. D. Watson, F. Mazzola, O. J. Clark, K. Underwood, I. Markovic, M. McLaren, A. Hunter, D. M. Burn, L. B. Duffy, S. Barua, G. Balakrishnan, F. Bertran, P. Le Fevre, T. K. Kim, G. van der Laan, T. Hesjedal, P. Wahl and P. D. C. King, *Nano Letters*, 2018, **18**, 4493–4499.
- 17 G. Duvjir, B. K. Choi, I. Jang, S. Ulstrup, S. Kang, T. Thi Ly, S. Kim, Y. H. Choi, C. Jozwiak, A. Bostwick, E. Rotenberg, J.-G. Park, R. Sankar, K.-S. Kim, J. Kim and Y. J. Chang, *Nano Letters*, 2018, **18**, 5432–5438.
- 18 F. Li, K. Tu and Z. Chen, *The Journal of Physical Chemistry C*, 2014, **118**, 21264–21274.
- 19 A. O. Fumega and V. Pardo, *arXiv e-prints*, 2018, arXiv:1804.07102.
- 20 I. A. Gospodarev, A. V. Eremenko, T. V. Ignatova, G. V. Kamarchuk, I. G. Kolobov, P. A. Minaev, E. S. Syrkin, S. B. Feodosyev, V. D. Fil, A. Soreau-Leblanc, P. Molinie and E. C. Faulques, *Low Temperature Physics*, 2003, **29**, 151–154.
- 21 F. Ersan, S. Cahangirov, G. Gökoğlu, A. Rubio and E. Aktürk, *Phys. Rev. B*, 2016, **94**, 155415.
- 22 V. V. Mazurenko, M. V. Valentyuk, R. Stern and A. A. Tsirlin, *Phys. Rev. Lett.*, 2014, **112**, 107202.
- 23 G. Beutier, S. P. Collins, O. V. Dimitrova, V. E. Dmitrienko, M. I. Katsnelson, Y. O. Kvashnin, A. I. Lichtenstein, V. V. Mazurenko, A. G. A. Nisbet, E. N. Ovchinnikova and D. Pincini, *Phys. Rev. Lett.*, 2017, **119**, 167201.
- 24 D. Pincini, F. Fabrizio, G. Beutier, G. Nisbet, H. Elnaggar, V. E. Dmitrienko, M. I. Katsnelson, Y. O. Kvashnin, A. I. Lichtenstein, V. V. Mazurenko, E. N. Ovchinnikova, O. V. Dimitrova and S. P. Collins, *Phys. Rev. B*, 2018, **98**, 104424.
- 25 I. B. Bersuker, *Journal of Physics: Conference Series*, 2017, **833**, 012001.
- 26 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 27 G. Kresse and J. Furthmüller, *Phys. Rev. B*, 1996, **54**, 11169–11186.
- 28 G. Kresse and J. Hafner, *Phys. Rev. B*, 1993, **47**, 558–561.
- 29 E. Vatansever, S. Sarikurt and R. F. L. Evans, *Materials Research Express*, 2018, **5**, 046108.
- 30 S. Grimme, *Journal of Computational Chemistry*, 2004, **25**, 1463–1473.
- 31 F. Ersan, H. D. Ozaydin and O. Üzengi Aktürk, *Philosophical Magazine*, 2019, **99**, 376–385.
- 32 D. Kaltsas and L. Tsetseris, *Journal of Physics: Condensed Matter*, 2017, **29**, 085702.
- 33 J. Wilson and A. Yoffe, *Advances in Physics*, 1969, **18**, 193–335.
- 34 Y. Ma, Y. Dai, M. Guo, C. Niu, Y. Zhu and B. Huang, *ACS Nano*, 2012, **6**, 1695–1701.
- 35 D. W. Boukhvalov, D. R. Dreyer, C. W. Bielawski and Y.-W. Son, *ChemCatChem*, 2012, **4**, 1844–1849.
- 36 I. V. Kashin, S. N. Andreev and V. V. Mazurenko, *Journal of Magnetism and Magnetic Materials*, 2018, **467**, 58 – 63.
- 37 A. I. Liechtenstein, M. I. Katsnelson and V. A. Gubanov, *Journal of Physics F: Metal Physics*, 1984, **14**, L125–L128.
- 38 G. Duvjir, B. K. Choi, I. Jang, S. Ulstrup, S. Kang, T. Thi Ly, S. Kim, Y. H. Choi, C. Jozwiak, A. Bostwick, E. Rotenberg, J.-G. Park, R. Sankar, K.-S. Kim, J. Kim and Y. J. Chang, *Nano Letters*, 2018, **18**, 5432–5438.
- 39 A. Togo and I. Tanaka, *Scr. Mater.*, 2015, **108**, 1–5.
- 40 G. V. Kamarchuk, A. V. Khotkevich, V. M. Bagatsky, V. G. Ivanov, P. Molinié, A. Leblanc and E. Faulques, *Phys. Rev. B*, 2001, **63**, 073107.

Supplementary Information: Structural phase transitions in VSe₂: energetics, electronic structure and magnetism[†]

Georgy V. Pushkarev,^a Vladimir G. Mazurenko,^a Vladimir V. Mazurenko^a and Danil W. Boukhvalov^{*a,b}

Fig. S1 shows angle dependencies of the total energy and magnetic moment in the case of the rotation of the whole Se upper layer of VSe₂ monolayer (a,b) and bulk (c,d) within the plane scheme.

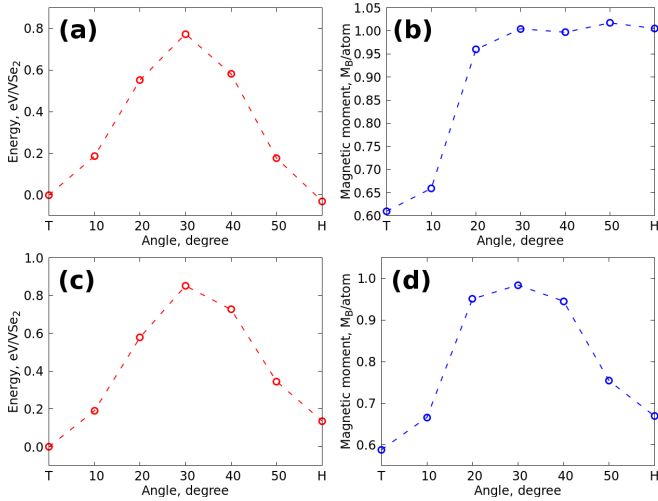


Fig. S1 Total energy (a,c) and magnetic moment (b,d) as functions of the rotation angles obtained within the plane model for all Se atoms belonging to the upper layer. The simulations were performed for VSe₂ monolayer (a,b) and bulk (c,d).

For monolayer one can see that such a rotation scheme is less profitable in energy, since the barrier grows. In turn, the magnetic moment demonstrates the same behavior as with the arc rotation model. In the bulk case we obtain almost the same dependencies, but the maximum of the energy barrier at 30° becomes larger than that in the case of the monolayer.

Also we calculated partial densities of states of VSe₂ bulk in intermediate points of arc type of rotation with the 20° step (Fig S2).

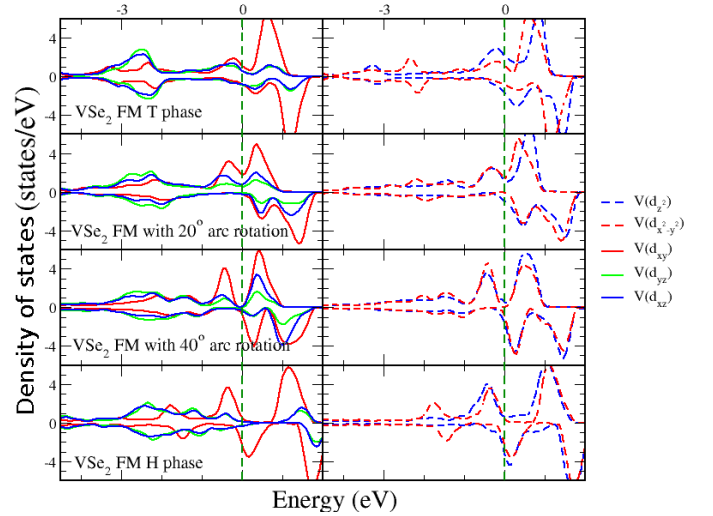


Fig. S2 Partial densities of states calculated for VSe₂ bulk in the ferromagnetic configuration. The arc rotation scheme with the 20° step was used. Left and right panels correspond to (d_{xy}, d_{yz}, d_{xz}) and $(d_{x^2-y^2}, d_{z^2})$ sets of states, respectively.

Figures S3 and Fig. S4 give band structures of VSe₂ monolayer and bulk obtained within arc scheme rotation with the 10° elementary step.

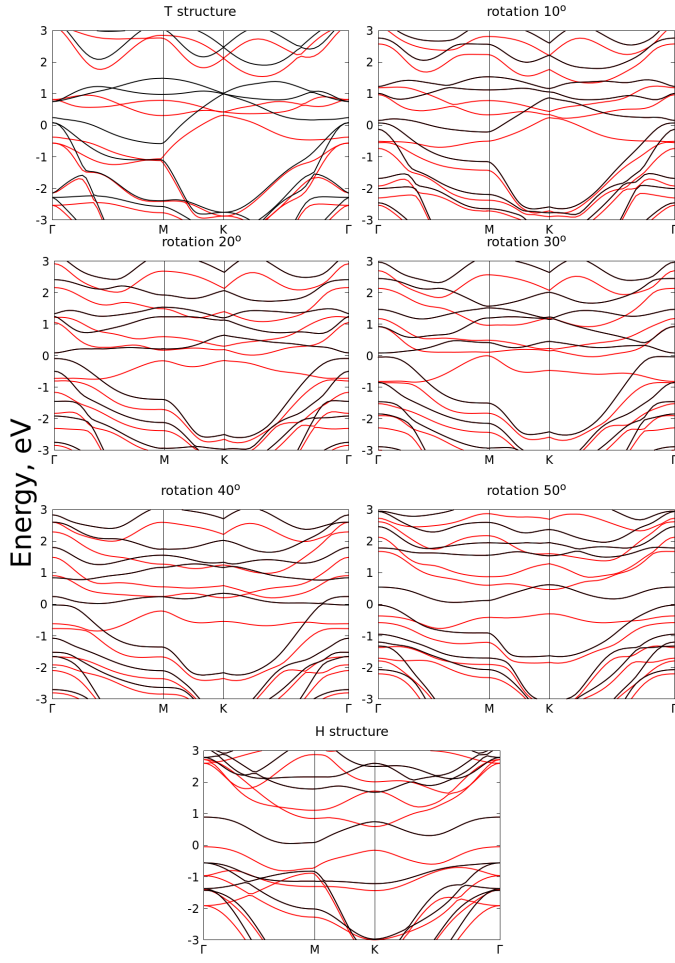


Fig. S3 Band structures of the monolayer VSe_2 calculated for atomic structures modified within the arc rotation model from T phase (0°) to H phase (60°) with the step of 10° . All the calculations were performed for ferromagnetic configuration. Red lines correspond to spin up states and black ones to spin down.

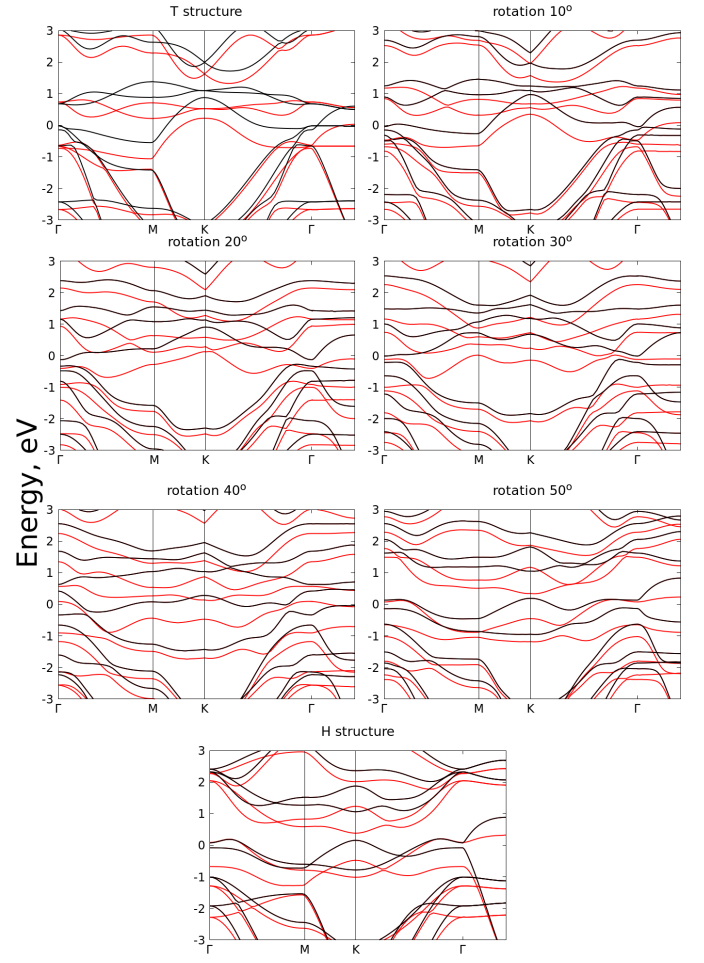


Fig. S4 Band structures of the bulk VSe_2 crystal calculated for atomic structures modified within the arc rotation model from T phase (0°) to H phase (60°) with the step of 10° . All the calculations were performed for ferromagnetic configuration. Red lines correspond to spin up states and black ones to spin down.